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# The Structure of Orthorhombic $\mathbf{Y}_{\mathbf{2}} \mathbf{T i O}_{5}$, an Example of Mixed Sevenand Fivefold Coordination 

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#### Abstract

Crystals of $\mathrm{Y}_{2} \mathrm{TiO}_{5}$, grown by heating the composition $\mathrm{NaYTiO}_{4}$ at $1500^{\circ} \mathrm{C}$ in air, have the unit-cell dimensions $a=10 \cdot 35, b=3 \cdot 70, c=11 \cdot 25 \AA$, space group Pnma, $Z=4$. The structure was deduced from the Patterson function and refined by three-dimensional Fourier and least-squares methods. The two independent yttrium ions are both in sevenfold coordination, with oxygen atoms forming irregular trigonal prisms capped through one rectangular face. These polyhedra unite by edge-sharing into groups of four extending infinitely in the direction of the short $b$ axis. The groups are held together by the titanium ions which are five-coordinated, with four oxygen ligands from one group and one from the next at the corners of a square pyramid. A number of rare-earth titanates with the same stoichiometry are isomorphous with $\mathrm{Y}_{2} \mathrm{TiO}_{s}$. The relationships to the $B$-type rare-earth oxide, to $\mathrm{YO}(\mathrm{OH})$ and to $\mathrm{Sr}_{2} \mathrm{PbO}_{4}$ structure types are discussed.


## Introduction

The mixed-metal oxycompounds formed at high temperature between Ti or Zr and the rare earths were summarized recently by Collongues, Queyroux, Perez y Jorba \& Gilles (1965). They reported a number of cubic phases $A_{2} B_{2} \mathrm{O}_{7}$ ( $A$ trivalent, $B$ quadrivalent) with the pyrochlore structure, a rhombohedral group $A_{4} B_{3} \mathrm{O}_{12}$ related to $\mathrm{UY}_{6} \mathrm{O}_{12}$ (Bartram, 1966) and to $\mathrm{Sc}_{4} \mathrm{Zr}_{3} \mathrm{O}_{12}$ (Thornber, Bevan \& Graham, 1968), as well as a third family $A_{2} B_{5}$ existing in both low- and high-temperature modifications. Each of the powder diffraction patterns of low-temperature $\mathrm{Y}_{2} \mathrm{TiO}_{5}, \quad \mathrm{Dy}_{2} \mathrm{TiO}_{5}$ and $\mathrm{Gd}_{2} \mathrm{TiO}_{5}$ was indexed in terms of a monoclinic superlattice $a \simeq 3 a^{\prime}, b \simeq a^{\prime}, c \simeq 3 a^{\prime}, \beta=90-\delta$, originating from a fluorite-type sub-cell of side $a^{\prime}$, and conforming to the notion of a phase intermediate in structure and composition between fluorite-type (stoichiometry $\mathrm{BO}_{2}$ ) and $C$-type rare-earth oxide $\left(A_{2} \mathrm{O}_{3}\right)$, both of which are cubic.

We now wish to report that a well-crystallized product, formed as one of the thermal decomposition
products of $\mathrm{NaYTiO}_{4}$, appears to be identical with the low-temperature form of $\mathrm{Y}_{2} \mathrm{TiO}_{5}$. This phase was overlooked by Ault \& Welch (1966) in their appraisal of the $\mathrm{Y}_{2} \mathrm{O}_{3}-\mathrm{TiO}_{2}$ system. The symmetry and unit-cell dimensions are not those given by Collongues et al. (1965), however, and the structure determination relates it more to the $B$ (monoclinic) rare-earth oxides than to the $C$ (cubic) form.

## Experimental

$\mathrm{NaYTiO}_{4}$, which is readily prepared by solid-state reaction between $\mathrm{Y}_{2} \mathrm{O}_{3}, \mathrm{TiO}_{2}$ and sodium oxalate at $1300^{\circ} \mathrm{C}$ in a sealed system, is related to a number of titanates with the $\mathrm{K}_{2} \mathrm{NiF}_{4}$ structure (Mumme, Reid \& Wadsley, 1967). It decomposes on heating in air at $1500^{\circ} \mathrm{C}$ into a heterogeneous product containing a number of small clear colourless needle-like crystals, which proved from this structure analysis to have the composition $\mathrm{Y}_{2} \mathrm{TiO}_{5}$. The crystallographic constants in Table 1 were determined from the separation of the $\alpha_{1}-\alpha_{2}$ doublet for a number of spots in the back-reflex-

Table 1. Crystallographic data for $\mathrm{Y}_{2} \mathrm{TiO}_{5}$

| Symmetry: orthorhombic |  |
| :---: | :---: |
| Unit-cell dimensions | $\begin{aligned} & a=10 \cdot 35 \pm 0 \cdot 01 \AA \\ & b=3.70 \pm 0.01 \\ & c=11 \cdot 25 \pm 0.01 \end{aligned}$ |
| Systematically absent reflexions | $0 k l$ with $k+l \neq 2 n$ |
|  | $h k 0$ with $h \neq 2 n$ |
| Possible space groups $D_{x}$ | Pnma (No.62), Pn2 $2_{1}$ (No.33) $4.73 \mathrm{~g} . \mathrm{cm}^{-3}$ |
| ${ }_{\mu}^{Z}$ | ${ }_{537 \mathrm{~cm}^{-1}}$ |

ion region of Weissenberg films by a simple variant of Main \& Woolfson's (1963) method.

The identity of this phase with the $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ of $\mathrm{Col}-$ longues et al. (1965) was established by a roundabout procedure. Waring \& Schneider (1965) prepared $\mathrm{Sm}_{2} \mathrm{TiO}_{5}, \mathrm{Eu}_{2} \mathrm{TiO}_{5}$ and $\mathrm{Gd}_{2} \mathrm{TiO}_{5}$, noted the similarity of the powder patterns, and yet were unable to confirm the unit-cell assignment of Collongues et al. They kindly sent a specimen of $\mathrm{Gd}_{2} \mathrm{TiO}_{5}$ to us, and a Guinier pattern of it was readily indexed in terms of an orthorhombic unit cell with the dimensions $a=10 \cdot 48, b=$ $3.76, c=11.33 \AA$, the missing reflexions being characteristic of the space group Pnma. $\mathrm{Gd}_{2} \mathrm{TiO}_{5}$ is clearly isostructural with the crystals prepared by thermally decomposing $\mathrm{NaYTiO}_{4}$ (Table 1) and both are members of a group of titanates $\mathrm{M}_{2} \mathrm{TiO}_{5}$ containing a number of the rare earths, M .
Intensities were collected with a crystal 0.08 mm long (corresponding to the $\mathbf{b}$ direction) and between 0.02 and 0.03 mm in average cross-section. $\mu r$ was in the range $1 \cdot 2-1 \cdot 4$ and absorption corrections were unnecessary. Exposures for the levels $h k l, k=0,1$ and 2 were made on multiple films with filtered copper radiation using an integrating Weissenberg camera, and measurements were made visually with a standardized scale of timed exposures having the same spot shape. Approximate scaling between levels was made initially by Wilson's (1942) method, and adjusted in between refinement cycles by comparing $F_{o}$ with $F_{c}$. Scattering curves for $\mathrm{Y}^{3+}$ and $\mathrm{Ti}^{4+}$, corrected with Cromer's (1965) values for $\Delta f^{\prime}$ and $\Delta f^{\prime \prime}$ were taken from Cromer \& Waber (1965), and the values for doubly ionized oxygen from Suzuki (1960). The weighting scheme for the least-squares refinement was that of Cruickshank, Pilling, Bujosa, Lovell \& Truter (1961), and the matrices were inverted with the block-diagonal approximation. All calculations were made on the Division's Elliott 803 computer using the programming system of Daly, Stephens \& Wheatley (1963).
The principal structural features were determined in projection to ( 010 ), as no extensive overlap of atoms is possible with a projection axis only $3 \cdot 70 \AA$ long. The derivation of the structure depended upon fixing the positions of two independent yttrium atoms, one titanium and five oxygen atoms in the asymmetric unit, all of which were assumed to lie in the mirror planes
at $y=\frac{1}{4}$ or $y=\frac{3}{4}$. Only one set of metals consistent with the Patterson projection $P(u w)$ were found, and their coordinates were improved and the five oxygen atoms located directly and without assumptions by reiteration of electron density and structure factor calculations based on the $h 0 l$ data. Preliminary bond length calculations were helpful in assigning the $y$ parameters for


Fig. 1. Superimposed electron density sections at $y=\frac{1}{4}$ and $\frac{3}{4}$. Contours drawn at intervals of $2 \cdot 5$ electrons for Ti and $\mathbf{O}$, and 5 electrons for the two yttrium atoms. Zero contour dotted.


Fig.2. Structure of $\mathrm{Y}_{2} \mathrm{TiO}_{5}$. Oxygen atoms are the largest circles, yttrium atoms are stippled and titanium atoms are the smallest circles, open and black. Heavier outlines or shading represents an atom at $y=\frac{3}{4}$, lighter at $y=\frac{1}{4}$. The bonds between titanium and oxygen are drawn as dashed lines.
all atoms. These were confirmed by three-dimensional Fourier sections coinciding with the mirror planes (Fig.1), and the positions were refined together with individual isotropic temperature factors, by a number of least-squares cycles using all of the data except for five reflexions of low indices, severely affected by extinction, which were omitted. Refinement was stopped when the shift of each variable was less than one quarter of its standard deviation. No attempt was made either to reduce the symmetry to $P n 2_{1} a$ or to correct for extinction. The final $R$ value of $12 \cdot 2 \%$ was rather high, but a three-dimensional difference synthesis con-
tained no peak or trough greater than $0 \cdot 5 \mathrm{e} . \AA^{-3}$. A comparison between $F_{o}$ and $F_{c}$ is given in Table 2, and the atomic parameters, the temperature factors, and their e.s.d.'s in Table 3. The structure is illustrated by Fig. 2.

## Discussion

Both yttrium atoms are seven-coordinated to oxygen, but the polyhedra, which are similar to each other, are not easy to define. One description is a cube of oxygen atoms from which one is missing and with some relaxation of three of the remaining seven [Fig.3(a)]. In


Fig. 3. Coordination of yttrium in $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ : (a) as a cube with one oxygen atom missing and with some rearrangement of the remainder; $(b)$ as a distorted trigonal prism with a seventh oxygen atom bonded through a face; (c) as an octahedron with one additional oxygen atom.

Table 3. Fractional atomic coordinates for $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ Space group Pnma (No.62). All atoms in point position 4(c). E.s.d.'s are given in brackets.

|  | $x$ | $y$ | $z$ | $B$ |
| :--- | :--- | :--- | :--- | :--- |
|  |  |  |  |  |
| $\mathrm{Y}(1)$ | $0.1156(3)$ | $\frac{1}{4}$ | $0.2231(3)$ | $0.51(6) \AA^{2}$ |
| $\mathrm{Y}(2)$ | $0.1366(3)$ | $\frac{4}{4}$ | $0.5578(3)$ | $0.43(6)$ |
| Ti | $0.1745(8)$ | $\frac{4}{4}$ | $0.8806(7)$ | $0.72(13)$ |
| $\mathrm{O}(1)$ | $0.4947(32)$ | 4 | $0.1024(29)$ | $1.2(6)$ |
| $\mathrm{O}(2)$ | $0.2229(31)$ | $\frac{4}{4}$ | $0.0449(28)$ | $0.9(6)$ |
| $\mathrm{O}(2)$ | $0.2594(30)$ | $\frac{4}{2}$ | $0.7340(27)$ | $0.9(6)$ |
| $\mathrm{O}(4)$ | $0.5085(28)$ | 4 | $0.6601(26)$ | $0.8(5)$ |
| $\mathrm{O}(5)$ | $0.2690(34)$ | $\frac{1}{4}$ | $0.3833(30)$ | $1.4(7)$ |

Table 2. Observed and calculated structure factors

this way the coordination resembles that of Zr in $\mathrm{ZrO}_{2}$ (McCullough \& Trueblood, 1959). Another is to relate the polyhedron to a trigonal prism with a seventh oxygen bonded to the metal through the centre of one rectangular face [Fig. $3(b)$ ], and this is the configuration we have chosen. A third alternative [Fig. 3(c)] shows the resemblance to an irregular octahedron with the addition of a seventh atom. The yttrium-oxygen distances for $Y(1)$ vary between 2.29 and $2.40 \AA$ and for $Y(2)$ between 2.32 and $2.38 \AA$, all $\pm 0.03 \AA$ (Table 4). Coordination numbers of 6 (octahedral), 7, 8 and 9 have been recorded for yttrium in oxy-, hydroxy- and oxyhalide compounds, and a comparison of bond lengths for a representative collection from the recent literature is given in Table 5, the Y-O bond increasing with coordination number. The 7-, 8-, and 9 -coordi-
nated polyhedra are most conveniently represented as the additions of seventh, eighth and ninth atoms to one or more of the three rectangular faces of a trigonal prism.

The irregularity of the yttrium environments in $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ may be attributed in part to the proximity of the titanium atom. This has fivefold coordination. Four of its neighbours $\mathrm{O}(2), \mathrm{O}(3)$ and $\mathrm{O}(5)$ (twice) are square planar and are already bonded to $Y(1)$ and $\mathrm{Y}(2)$ from a double group consisting of the two polyhedra sharing edges. The fifth, $\mathrm{O}(4)$, is from the next double group and forms the apex of a square pyramid of oxygen atoms. Titanium is displaced by about $0.05 \AA$ from the centre of the square base towards this apex, and the $\mathrm{Ti}-\mathrm{O}(4)$ distance of $1.78 \AA$ is the shortest of the five (Table 4). There is no sixth atom closer to

Table 4. Interatomic distances $(\AA)$
The e.s.d.'s for $\mathrm{Y}-\mathrm{O}$ and for $\mathrm{Ti}-\mathrm{O}$ are all $\pm 0.03 \AA$, and $\pm 0.04 \AA$ for $\mathrm{O}-\mathrm{O}$. The figures in brackets following the bond lengths are the numbers of these distances within a single coordination polyhedron.

| $\mathrm{Y}(1)-\mathrm{O}(1)$ | 2.33 (1) | $\mathrm{Y}(2)-\mathrm{O}(1)$ | 2.32 (1) | $\mathrm{Ti}-\mathrm{O}(2)$ | 1.91 (1) |
| :---: | :---: | :---: | :---: | :---: | :---: |
| -O(2) | $2 \cdot 29$ (1) | -O(3) | 2.35 (1) | -O(3) | 1.87 (1) |
| -O(5) | $2 \cdot 40$ (1) | -O(5) | 2.38 (1) | -O(4) | 1.78 (1) |
| -O(3) | $2 \cdot 26$ (2) | -O(1') | 2.35 (2) | -O(5) | 1.94 (2) |
| -O(4) | $2 \cdot 36$ (2) | -O(2) | 2.35 (2) | $\mathrm{O}(5)-\mathrm{O}(2)$ | $2 \cdot 58$ (2) |
| $\mathrm{O}(2)-\mathrm{O}(4)$ | 3.29 (2) | $\mathrm{O}(3)-\mathrm{O}(2)$ | 2.82 (2) | -O(3) | 2.52 (2) |
| -O(3) | $2 \cdot 82$ (2) | -O(1) | $3 \cdot 54$ (2) | -O(4') | $2 \cdot 99$ (2) |
| $\mathrm{O}(3)-\mathrm{O}(4)$ | 2.71 (2) | $\mathrm{O}(2)-\mathrm{O}\left(1^{\prime}\right)$ | 2.89 (2) | $\mathrm{O}\left(2^{\prime}\right)-\mathrm{O}\left(4^{\prime}\right)$ | $3 \cdot 20$ (2) |
| -O(5) | $2 \cdot 52$ (2) | -O(5) | 2.58 (2) | $\mathrm{O}(3)-\mathrm{O}\left(4^{\prime}\right)$ | $2 \cdot 86$ (2) |
| $\mathrm{O}(1)-\mathrm{O}(4)$ | $3 \cdot 24$ (2) | $\mathrm{O}\left(1^{\prime}\right)-\mathrm{O}\left(1^{\prime \prime}\right)$ | 2.95 (2) |  |  |
| -O(5) | $2 \cdot 84$ (1) | $\mathrm{O}\left(5^{\prime}\right)-\mathrm{O}\left(1^{\prime \prime}\right)$ | $2 \cdot 84$ (1) |  |  |
| $\mathrm{O}(3)-\mathrm{O}(3)$ | 3.70** | $\mathrm{O}(2)-\mathrm{O}(2)$ | $3 \cdot 70^{*}$ |  |  |
| $\mathrm{O}(4)-\mathrm{O}(4)$ | 3.70* | $\mathrm{O}(1)-\mathrm{O}(1)$ | 3.70* |  |  |

Table 5. Yttrium-oxygen distances

| Compound | Coordination number | $\mathrm{Y}-\mathrm{O}$ distances |  | Reference |
| :---: | :---: | :---: | :---: | :---: |
|  |  | Range | Average |  |
| $\mathrm{Y}_{2} \mathrm{O}_{3}$ | 6 | 2.26-2.35 $\AA$ | 2.30 A | Paton \& Maslen (1965) |
| $\mathrm{Y}_{2} \mathrm{BeO}_{4}$ | 6 | 2.21-2.34 | $2 \cdot 29$ | Harris \& Yakel (1967) |
| $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ | 7 | 2-29-2.40 | 2.32 | This study Y(1) |
|  |  | 2.32-2.38 | $2 \cdot 35$ | This study Y(2) |
| $\mathrm{YO}(\mathrm{OH})$ | 7 | 2.24-2.44 | 2.33 | Christensen (1965) |
| $\mathrm{Y}_{3} \mathrm{Fe}_{5} \mathrm{O}_{12}$ | 8 | 2.37-2.43 | $2 \cdot 40$ | Geller \& Gilleo (1957) |
| $\mathrm{Y}(\mathrm{OH})_{2} \mathrm{Cl}$ | $\begin{gathered} 8 \\ (6 \mathrm{O}+2 \mathrm{Cl}) \end{gathered}$ | 2.31-2.39 | $2 \cdot 36$ | Klevtsova \& Klevtsov (1966) |
| $\mathrm{Y}(\mathrm{OH})_{3}$ | 9 | 2.40-2.44 | $2 \cdot 42$ | Christensen, Hazell \& Nilsson (1967) |

Table 6. Five-coordinated $\mathrm{Ti}-\mathrm{O}$ distances

| Compound | Ti-O distances |  | Reference |
| :---: | :---: | :---: | :---: |
|  | Range | Average |  |
| $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ | 1.78-1.94 | 1.89 £ | This study |
| $\mathrm{K}_{2} \mathrm{Ti}_{2} \mathrm{O}_{5}$ | 1.57-2.00 | 1.85 | Andersson \& Wadsley (1961) |
| $\mathrm{Ba}_{2} \mathrm{TiSi}_{2} \mathrm{O}_{8}$ | 1.66-2.00 | 1.93 | Moore \& Louisnathan (1967) |
| Dichlorodiphenoxytitanium(IV) | $\begin{aligned} & 1 \cdot 74-2 \cdot 12 \\ & \text { (oxygen only) } \end{aligned}$ |  | Watenpaugh \& Caughlan (1966) |
| (3 oxygen, 2 chlorine ligands) |  |  |  |




Fig.4. (below) Two ways of joining seven-coordinated polyhedra by a common edge. (above) The same as atoms, the smaller being the metal and the larger the oxygen atoms.

(b)

Fig. 5. (a) and (b) Both units of Fig. 4 extended indefinitely: (above) in perspective; (below) in projection down the repeat direction.

Ti than $\mathrm{O}(\mathrm{l})$, which is $3.89 \AA$ away and is clearly unbonded.

Fivefold coordination for $\mathrm{Ti}^{4+}$ can no longer be considered a rarity. It was first found by Andersson \& Wadsley (1961) in the hygroscopic layer compound $\mathrm{K}_{2} \mathrm{Ti}_{2} \mathrm{O}_{5}$, and has subsequently been identified in the ester dichlorodiphenoxytitanium(IV) (Watenpaugh \& Caughlan, 1966) as well as in two minerals, fresnoite $\mathrm{Ba}_{2} \mathrm{TiSi}_{2} \mathrm{O}_{8}$ (Moore \& Louisnathan, 1967; Masse, Grenier \& Durif, 1967) and lamprophyllite ( $\mathrm{Ba}, \mathrm{Sr}, \mathrm{K}$ )$\mathrm{Na}(\mathrm{Ti}, \mathrm{Fe}) \mathrm{TiSi}_{2} \mathrm{O}_{7}(\mathrm{O}, \mathrm{OH}, \mathrm{F})_{2}$ (Woodrow, 1964). $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ as well as these minerals is very stable, and evidently the reactive nature of the other two compounds is not simply due to an unstable coordination of Ti , as might at first be supposed. $\mathrm{Ti}-\mathrm{O}$ distances are given in Table 6 for all of these compounds except lamprophyllite, its structure so far having been determined only in one projection. In this mineral, in fresnoite and in $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ the coordination is square pyramidal, and in all cases the shortest distance is from the metal to the oxygen at the apex. There are two short $\mathrm{Ti}-\mathrm{O}$ bond lengths of 1.57 and $1.67 \AA$ in $\mathrm{K}_{2} \mathrm{Ti}_{2} \mathrm{O}_{5}$ where the coordination of Ti , and also in the case of the ester, is described as trigonal bipyramidal. An unambiguous decision between these two modes is not always practicable or meaningful, however, and may simply be a matter of semantics.

## Sevenfold coordination

There is a good deal of growing evidence that metaloxygen octahedra tend to aggregate by edge-sharing into groups extending infinitely through a crystal in one and sometimes two directions. These groups, or structural building blocks, are able to unite with themselves through common corners or edges, to form three-dimensional structures which are restricted in number by the configuration of the group, as well as by the presence of additional ions of specific sizes within the interstices between the groups (Wadsley, 1967). It may therefore be worth while to see if these simplifying working 'rules' can be extended to the trigonal prism having a seventh atom additionally bonded through a rectangular face.

While it would be a useful exercise to list exhaustively all possible ways of joining these figures together, as Aurivillius \& Lundgren (1965) have recently done for the 8 -coordinated square Archimedean antiprism, we confine this discussion to a number of structures already determined in order to keep it within reasonable bounds. These include $\mathrm{Y}(\mathrm{OH})_{3}$ (Schubert \& Seitz, 1946; Christensen, Hazell \& Nilsson, 1967), and $\mathrm{YO}(\mathrm{OH})$ (Klevtsova \& Klevtsov, 1964; Christensen, 1965), high-pressure $\mathrm{Y}_{2} \mathrm{O}_{3}$ (Hoekstra \& Gingerich, 1964 ) with the $B$-type rare-earth oxide structure determined for $\mathrm{Sm}_{2} \mathrm{O}_{3}$ by Cromer (1957), and also the phases isostructural with $\mathrm{Sr}_{2} \mathrm{PbO}_{4}$ including $\mathrm{Ca}_{2} \mathrm{PbO}_{4}, \mathrm{Ca}_{2} \mathrm{SnO}_{4}$ (Trömel, 1965, 1967) and $\mathrm{Mn}_{2} \mathrm{GeO}_{4}$ (Wadsley, Reid \& Ringwood, 1968) but for which no yttrium isotype has yet been found.

We can extract two features which are common to these four structures and also to $\mathrm{Y}_{2} \mathrm{TiO}_{5}$. These are illustrated in Fig.4, showing two of a number of possible ways of joining the seven-coordinated polyhedra by common edges. These two units unite with themselves to form groups of two different kinds, both extending indefinitely through a structure with a repeat distance equal to an oxygen-oxygen separation (Fig.5).
$\mathrm{Sr}_{2} \mathrm{PbO}_{4}$ and its isotypes contain only one kind of group [Fig.5(a)], joined together by edges. The Sr atoms are the seven-coordinated atoms (or more nearly six-coordinated), and the Pb atoms are situated in octahedral positions between the groups (Fig.6). Both groups of Fig. 5 form the basic structure of the $B$-type rare-earth oxide (Fig.7). One-third of the metals are in the octahedral positions between the groups, and their coordination number is therefore six, and not seven, as often stated (Structure Reports, 1957).

The groups are also able to form much larger composite units. Fig. 8, containing both of the elements of Fig. 5, is the fundamental structural unit of $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ and its isotypes, which, as we have already seen, is grossly distorted by the presence of the Ti atoms binding them together in a way so as to conserve space (Fig.9).

Fig.5(b) can also be extended in a second direction to become an infinite two-dimensional sheet (Fig. 10). The structure of $\mathrm{YO}(\mathrm{OH})$ consists of these sheets arranged in parallel and bonded through two oxygen atoms and by hydrogen ions (Fig.11). The similarity between $\mathrm{YO}(\mathrm{OH})$ and $\mathrm{Y}(\mathrm{OH})_{3}$, where the yttrium is 9 -coordinated - having two more oxygen atoms beyond the two remaining rectangular faces - has been discussed by Klevtsova \& Klevtsov (1964) for the dehydration reaction

$$
\mathrm{Y}(\mathrm{OH})_{3} \rightarrow \mathrm{YO}(\mathrm{OH})+\mathrm{H}_{2} \mathrm{O} .
$$

It is clear to us that by no means all of the structures which can be generated by the close union of these polyhedra have been found. This may be due to the difficulty of synthesizing compounds containing the metals - apart from sodium - which are able to assume this particular coordination, $\mathrm{Sr}, \mathrm{Ca}, \mathrm{Mn}^{2+}, \mathrm{Y}, \mathrm{Gd}, \mathrm{Dy}$ and presumably more of the rare earths. All of these can be expected to form mixed oxide phases only at very high temperatures, and the difficulties of making crystals large enough for unambiguous X-ray studies can be overcome with normal laboratory equipment only by tricks of synthesis such as we have used in the present case. Finally the similarity between the environment of Y and Zr in oxides, mentioned earlier, suggests a close study of the relations of the structures in Figs. 6, 7, 8 and 11 to those of pyrochlore $\left(\mathrm{Y}_{2} \mathrm{Ti}_{2} \mathrm{O}_{7}\right.$ or $\mathrm{Y}_{2} \mathrm{Zr}_{2} \mathrm{O}_{7}$ ), where the coordination is a mixture of six and eight, and $\mathrm{ZrO}_{2}$ where it is seven.

We are grateful to Dr A.F. Reid for preparing the crystals.


Fig. 6. $\mathrm{Sr}_{2} \mathrm{PbO}_{4}$ structure type, consisting of the groups of Fig. 5(a), centred on the Sr atoms, joined by edges leaving octahedral holes for Pb (circles). One group is more heavily stippled.


Fig. 7. Structure of $\mathrm{Sm}_{2} \mathrm{O}_{3}$ (or of high-pressure $\mathrm{Y}_{2} \mathrm{O}_{3}$ ), the $B$-rare-earth-oxide type. The more heavily stippled groups emphasize two different varieties in Fig.5, and the circles are the metals in an octahedral environment.


Fig. 8. Composite of both units of Fig. 5; (above) in perspective, and (below) in projection down the repeat direction.

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Fig. 9. Structure of $\mathrm{Y}_{2} \mathrm{TiO}_{5}$ containing the composite groups of Fig. 8, considerably distorted. The Ti atoms are drawn as circles, with their bonds to oxygen as dashed lines. This Figure is rotated through $90^{\circ}$ with respect to Fig. 2 in order to conform to Fig. 8.


Fig.10. Formation of a two-dimensional infinite sheet from the groups in Fig. 5(b). (left) in projection; (right) in perspective.


Fig. 11. Structure of $\mathrm{YO}(\mathrm{OH})$, made up from the union of the sheets in Fig. 10 joined by edges. Hydrogen atoms are presumably located in the holes.

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